

THE SWELLING OF Ge-TAENIOLITE SERIES MICAS WITH WATER

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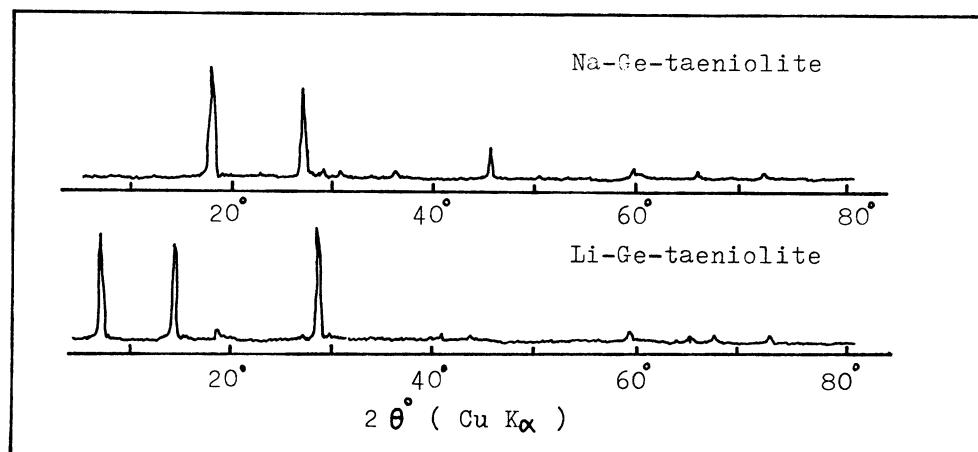
Germanium substituted analogues of Na-taeniolite and Li-taeniolite, i.e. , Na-Ge-taeniolite  $\text{NaMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2$  and Li-Ge-taeniolite  $\text{LiMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2$  were found to be expanded by water . However , the hydration tendencies of these Ge-taeniolites were found to be somewhat weaker than those of the corresponding silicon substituted analogues .

Although detailed investigations on the swelling of Na-taeniolite  $\text{NaMg}_2\text{Li}(\text{Si}_4\text{O}_{10})\text{F}_2$  1) and Li-taeniolite  $\text{LiMg}_2\text{Li}(\text{Si}_4\text{O}_{10})\text{F}_2$  2) were reported by the authors previously , studies on the corresponding germanium substituted analogues of them , i.e. , Na-Ge-taeniolite  $\text{NaMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2$  and Li-Ge-taeniolite  $\text{LiMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2$  , have been lacking . Recently we could substantially synthesize both these germanium taeniolites , and investigated the water uptake and dehydration processes of them extensively , using high temperature X-ray diffraction , TG and DTA methods . Crystallochemical similarity between these silicates and germanates was observed as was expected . The hydration tendencies of these two series micas were also compared .

Mixtures of 1 mol-NaF , 1 mol-LiF , 2 mol-MgO , and 4 mol- $\text{GeO}_2$  , which were special grade reagents , were melted in a sealed platinum crucible at  $1250^\circ\text{C}$  for 5 hrs , and then crystallized into Na-Ge-taeniolite at the cooling rate of  $3^\circ\text{C}/\text{min}$  . Li-Ge-taeniolite was also synthesized by the same method from the corresponding mixtures , which consisted of 2 mol-LiF , 2 mol-MgO and 4 mol- $\text{GeO}_2$  . The crystal aggregates of Na-Ge-taeniolite thus obtained were powdered , and purified by the sedimentation in distilled water . On the other hand the crystal aggregates of Li-Ge-taeniolite , which were immersed into distilled water , were bloated into a stable suspension , and then purified by centrifugation . The samples thus purified were dried at  $130^\circ\text{C}$  and then studied by the X-ray diffraction method . The X-ray diffraction patterns for Na-Ge-taeniolite and Li-Ge-taeniolite are shown in Fig.1 . The formation of both Ge-taeniolites was also ascertained by chemical analysis.

Fig. 2 shows the changes of X-ray diffraction patterns in the (001) reflection region of Na-Ge-taeniolite caused by the water uptake into the interlayer sites , when the samples are wetted by distilled water . The hydrated state A in Fig. 2 seems to be a so-called one-layer hydrated type , because the clearance space , i.e. , the difference between the interlayer distance  $d(001)$  of anhydrous Na-Ge-taeniolite ( $9.97 \text{ \AA}$ ) and that of the hydrated one ( $12.68 \text{ \AA}$ ) , is  $2.71 \text{ \AA}$  , which is nearly equal to the OH-O distance ( $2.76 \text{ \AA}$ ) in ice crystal 3) .

Fig. 1 X-ray diffraction patterns of Na-Ge-taeniolite and Li-Ge-taeniolite .  
Li-Ge-taeniolite is already hydrated absorbing water vapour in air .



The crystalline state of Na-Ge-taeniolite corresponding to maximum swelling is one-layer type , and a higher hydrated type is never obtained , whereas that of Na-taeniolite is two-layer type . The interlayer distance of one-layer hydrated Na-Ge-taeniolite is distinctly larger than that ( 12.26 Å ) of one-layer hydrated Na-taeniolite , a fact to which the difference between ionic radius of  $\text{Si}^{4+}$  ( 0.40 Å ) and that of  $\text{Ge}^{4+}$  ( 0.50 Å ) may be responsible . This one-layer hydrated type of Na-Ge-taeniolite is unstable , and is easily dehydrated at room temperature and humidity .

Fig. 3 shows the changes of X-ray diffraction patterns in the (001)

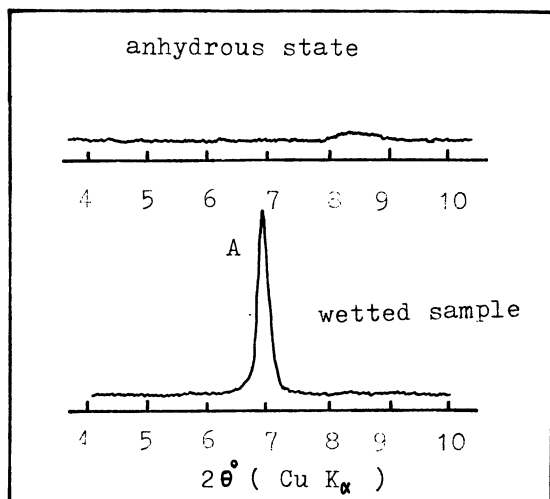


Fig.2 Changes of X-ray diffraction patterns in the (001) reflection region of Na-Ge-taeniolite caused by water uptake .

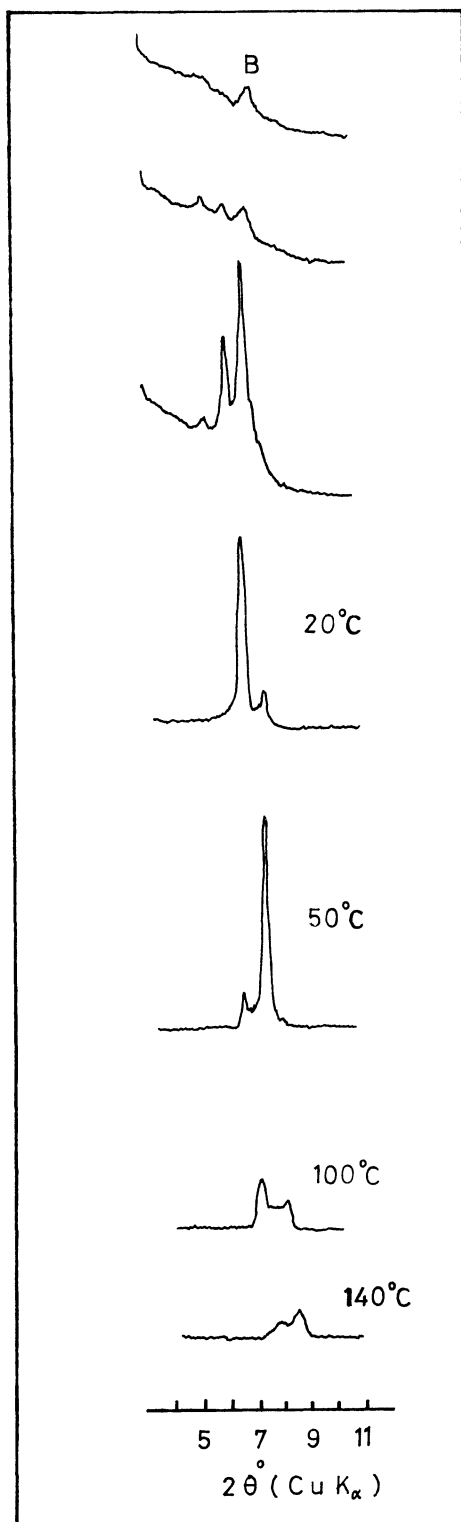
reflection region of Li-Ge-taeniolite caused by the decrease of water content .

Li-Ge-taeniolite gel showed no diffraction peak except a weak one at 14.2 Å ( B in Fig. 3 ) , whereas Li-taeniolite gel shows a weak one at 15.2 Å . It seems that the band scattering at the low diffraction angle in Fig.3 is caused by random water uptake into the interlayer sites of Li-Ge-taeniolite . It is obvious from this diffraction pattern that nearly free swelling occurs in Li-Ge-taeniolite - water system as in Li-taeniolite - water system .

The interlayer distances of (001) for predominant crystalline swelling states of Li-Ge-taeniolite , are 10.6 Å ( monohydrate state ) , 12.5 Å ( one-layer type ) , 14.2 Å ( trihydrate state ) , and 15.5 Å ( two-layer type ) , while those of the

corresponding states of Li-taeniolite are 10.4 Å , 12.2 Å , 13.9 Å , and 15.2 Å respectively . As shown before for Na-Ge-taeniolite and Na-taeniolite ,

Fig. 3 Changes of X-ray diffraction patterns in (001) reflection region caused by decreasing water content of Li-Ge-taeniolite .

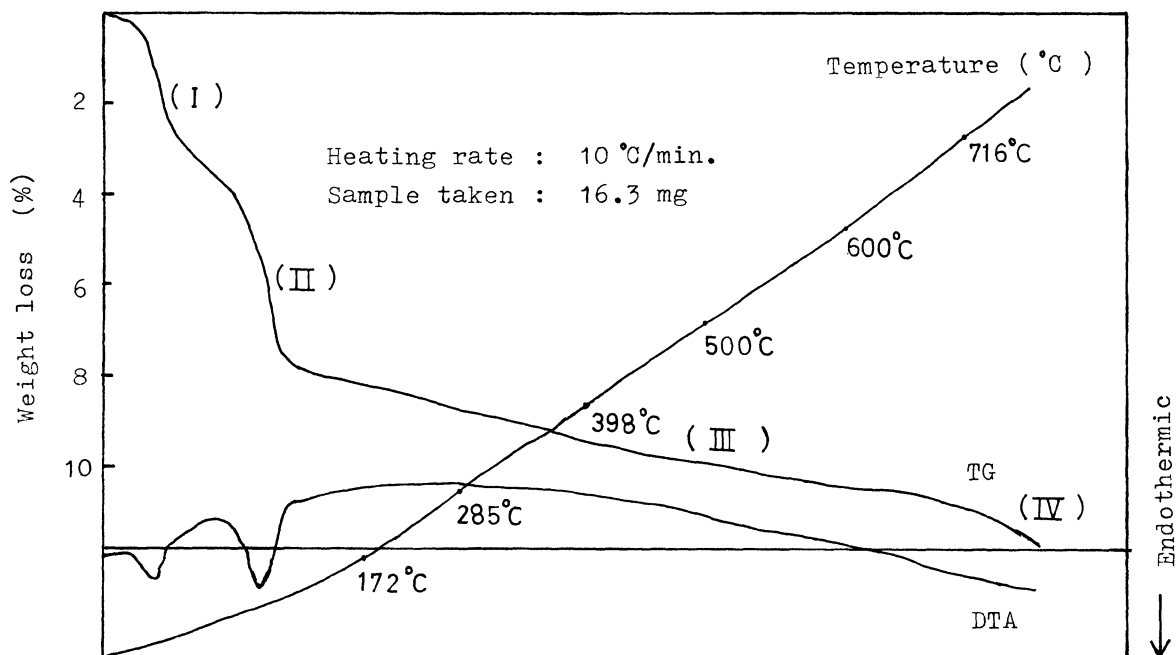


the interlayer distances of hydrated Li-Ge-taeniolites are also slightly longer than those of corresponding hydrated Li-taeniolites . The dehydration processes of hydrated Li-Ge-taeniolites are almost the same as those of hydrated Li-taeniolites , but the formers are dehydrated at slightly lower temperature than the latter . This increased dehydration tendency of hydrated Li-Ge-taeniolites is also demonstrated by the DTA endothermic peak temperature ; i.e. , the peak temperature is 123 °C for one-layer hydrated Li-Ge-taeniolite , which is lower than 136 °C for the same hydrated state of Li-taeniolite .

These slightly easier dehydration or weaker hydration tendencies of Ge-taeniolites in comparison with those of corresponding taeniolites may be explained by the difference between the configurational stability in the interlamellar region of Ge-taeniolites and that of taeniolites . The sodium ion and lithium ion can not establish contacts with surface oxygens of silicate or germanate layers as effectively as the potassium ion does , because of their small ionic radii . Thus there will be a close approach between interlayer  $\text{Li}^+$  or  $\text{Na}^+$  ion and  $\text{Si}^{4+}$  or  $\text{Ge}^{4+}$  ion , so that the repulsion between these cations will be considerable . But the degree of this repulsion will be substantially smaller for Ge-taeniolites than for taeniolites , because  $\text{Ge}^{4+}$  ion is larger than  $\text{Si}^{4+}$  ion , i.e. , interionic distances between  $\text{Ge}^{4+}$  and  $\text{Li}^+$  or  $\text{Na}^+$  ion in Ge-taeniolites are longer than those between  $\text{Si}^{4+}$  and  $\text{Li}^+$  or  $\text{Na}^+$  ion in taeniolites . So it will be reasonable to expect that the hydration of Ge-taeniolites will occur more mildly than that of taeniolites . It also seems that these effects , in combination with large hydration energies of interlayer  $\text{Li}^+$  and  $\text{Na}^+$  ions , are the predominant reasons why Ge-taeniolites can swell with water .

The TG curve for hydrated Li-Ge-taeniolite ( 14.2 Å phase ) was recorded by a thermobalance . It is obvious from Fig. 4 that the TG curve consists of four stages , the last one ( IV ) being due to the thermal decomposition of the sample . Thus the dehydration occurs in three stages .

Fig. 4 TG and DTA curves of trihydrated Li-Ge-taeniolite



The first stage (I) in the TG curve is due to the dehydration process from 14.2 Å phase to 12.5 Å phase, in accordance with the high temperature X-ray diffraction studies. The second stage (II) can also be ascribed to the dehydration process from 12.5 Å phase to 10.6 Å phase. The third stage (III) on the weight loss curve is then attributed to the dehydration process from 10.6 Å phase to anhydrous Li-Ge-taeniolite. The total weight loss attains 10.6%, which is nearly the same value as 9.80% of the calculated value of 14.2 Å phase (trihydrate state), chemical composition of which can be reasonably expressed as  $\text{LiMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2 \cdot 3\text{H}_2\text{O}$ . The weight losses of the second stage and the third one are 3.5% and 3.0%, respectively. From these results 12.5 Å phase and 10.6 Å phase can be expressed as  $\text{LiMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2 \cdot 2\text{H}_2\text{O}$  and  $\text{LiMg}_2\text{Li}(\text{Ge}_4\text{O}_{10})\text{F}_2 \cdot \text{H}_2\text{O}$ , respectively, because the theoretical value based on the chemical composition shown above is calculated to be 3.27% for each stage. The weight loss of hydrated Li-Ge-taeniolite is fairly small in comparison with hydrated Li-taeniolite, because of its larger molecular weight.

Thus the similarity and the differences between Ge-taeniolites and taeniolites are demonstrated. From the swelling characteristics, crystal structures and chemical compositions, hydrated Li-Ge-taeniolite and Na-Ge-taeniolite might be reasonably called as Ge-fluor-hectorite.

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